Structure and Physical Properties of 30MgSO$_4$-(70-x) P$_2$O$_5$-xSm$_2$O$_3$ glasses
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Abstract

Glasses in the series 30MgSO$_4$-(70-x) P$_2$O$_5$-xSm$_2$O$_3$, 0.2 ≤ x ≤ 1.0 mol percent were prepared by single step melt-quench techniques, the structure, physical and optical properties of samarium oxide at different concentrations were studied. The amorphous phase of the samples was verified by X-ray diffraction spectroscopy, the structural units were investigated using Fourier Transform Infrared, Raman and NMR spectroscopy where the spectra reveal the presence of symmetric, asymmetric, torsional and bending vibration of phosphate groups, the NMR spectra reveals the changes in Q$^n$ distribution here discussed, the compositional dependence on physical (density, average molar weight, dielectric constant, Inter-nuclear distance, Field strength, Molar volume, Molar refractivity etc) and optical parameters were determined. In terms of structural modifications, there is variation of density and molar volume, the glass sample has increasing refractive index values with decreasing molar volume content. The luminescence spectra were observed within UV-Vis-NIR region, the absorption spectra of the sample were recorded in the range of 350 to 1800 nm meanwhile, absorption edge the optical band-gap, and $\Delta E$ was evaluated. Our result may be advantageous for many optical applications (e.g. visible lasers or color displays).

Keywords: Melt-quench, Ultra-phosphate, Spectroscopy, Luminescence, Density

INTRODUCTION

With increasing demand of energy and the resulting outlays, there is a crucial need for the energy saving devices. Therefore, a consideration has to be given in many optical devices such as LEDs, solid state laser or any optical devices so as to replace/assist the conventional lightning. The glasses have wide range of applications in modern science and technologies among which phosphate based glasses has some advantages over the conventional silicate and borate glasses due their good glass forming ability [1], its posses high transparency, high gain density, low melting and softening temperature [2] with favorable solubility of rare earth ions [3] and used for verifications of certain nuclear wastes [4, 5]. The hygroscopic nature of the phosphate glasses was verified by the spectroscopic absorption bands due to hydroxyl groups been observed at certain range [6-9], the hygroscopic character leads to the weaker chemical stability and devitrification tendencies [10] which can’t be protected adequately, the addition of modifier oxides help to overcome these difficulties. Preparation of phosphate based glasses in the ultra- meta- and poly-phosphate region have been achieved [2, 11, 12] using some amounts of modifier oxides such as Na$_2$O [2] Ca-O [11] or Mg-O [12, 13]. Two competing modifier and intermediate oxides were combined in the glass system to modify and stabilized the phosphate structure with relatively lower amount. Even though, some report shows that the binary Magnesium phosphate (MgO-P$_2$O$_5$) glasses exhibit anomalous behavior near the Q$^2$ units [12, 13], this behavior attributed to change in physical properties of glasses which has been observed in relation to MgO coordination number. Furthermore, magnesium oxides can help in reducing the hygroscopic nature of host glasses [15] incorporation of sulfur results in structural modification of phosphate structure [16] as sulfur ions are widely dissolve in the P$_2$O$_5$ glass and mostly
remain as isolated units, for glasses with sulfo-phosphate ratio 1:2 the glass structure is entirely de-polymerized [17]. The spectroscopic and physical properties of tellurite, borate, silicate and phosphate glasses doped with various quantities of rare earth ions have been investigated previously[18, 19]. These ions incorporated with host materials increases their potential application in fabrication of optical fibers, memory devices, optoelectronics due to sufficient absorption and emission band obtained between the energy levels [20, 21]. Samarium oxides is among the most interesting rare earth ions doped glasses that exhibit a strong fluorescence within a visible region having a weaker concentration quenching and no up-conversion losses been used for laser materials [18] for hole burning studies [22] color display [23] and are fit to analyze the energy transfer processes. The present work is aimed to prepare and study the Sm$^{3+}$-doped magnesium sulfate ultra-phosphate glass with different samarium oxide concentration and its effect on physical and optical properties.

**Experimental**

**Samples preparation**

A series of Sm$^{3+}$-doped magnesium sulfate ultra-phosphate glasses of type 30MgSO$_{4}$-(70-x) P$_2$O$_5$-xSm$_2$O$_3$ with 0.2 ≤ x ≤ 1.0 mol % (referred to as MSPSm) were prepared by single step conventional melt quench method. The appreciable amount of reagent grade MgSO$_4$, H$_2$PO$_4$ and Sm$_2$O$_3$ weighed by digital electronic balance with a certified accuracy of ± 0.001 g of 99.8 % purity were mixed thoroughly for 20g batches. The samples preheated in an alumina crucible at 300 °C for 45 min, then transferred to electric furnace of higher temperature at 1200 °C for an hour, the samples was stirred occasionally to ensure homogenization and to obtained bubbles free. The samples were then poured directly on to a rectangular stainless steel plate and annealed for 5hours in order to eliminate the internal stress and preserved the melt in desiccators for further characterization. Samples of required optical quality were then selected, grinded and polished for spectroscopic properties, the chemical transformation equations of the substances preheated at 300 °C were indicated as:

- Magnesium sulfide: MgS + H$_2$SO$_4$ → MgSO$_4$ + H$_2$S ↑,
- Phosphoric acid: 2H$_3$PO$_4$ → P$_2$O$_5$ + 3H$_2$O ↑,

**Instrumentation**

**-X-ray measurement**

In the present study, the materials phase was determined using X-ray diffraction technique recorded at angle 2 theta in the range of 10°≤ θ ≤ 80° with scanning rate of 2.4°/min. The samples were examined with a diffraction software analysis of wavelength (Cu Kα, λ= 1.54056 Å) operating at 40 kV, 30 mA at room temperature. The broad hump centered at around 18-20° (=20) without any sharp band this verify the amorphous state of materials.

**-Density - molar volume measurement**

The density (ρ) of each sample were discovered by Archimedes and principle, the samples weighted in air and submerge in toluene of densities 0.865 g/cm$^3$, and are calculated using
where $W_a$ and $W_t$ is the weight of glass samples in air and that of toluene and $\rho_t$ is density of toluene, all sample weight measurements were made using a digital balance. The corresponding molar volume were obtained using the relation $V_m = M_t/\rho$ (cm$^3$/mol) as $M_t$ is the total molar weight of the samples and $\rho$ is the density in equation (1), the experiment was repeated four times and take the average values.

### Refractive index measurement

Refractive index ($n$) is the parameter used to determine the suitability for optical materials which are demonstrated using the subsequent equation (2) where $E_g$ is the optical band gap obtained from the absorption measurement.

$$\frac{n^2 - 1}{n^2 + 2} = 1 - \sqrt{\frac{E_g}{20}}$$

### Optical absorption measurement

The optical absorption spectra of bulk samples were determine in the range of 350-1800 nm (wave length) by double beam spectrophotometer at room temperature (Figure 7). The measurement made by Shimadzu 3101 PC UV-Vis-NIR Spectrophotometer located at Thin Film Laboratory in Physics Department of University Technology Malaysia. The molar absorption coefficient $\alpha(v)$ was evaluated using the Lambert-Beer law as [11, 24].

$$\alpha(v) = \left(\frac{1}{D}\right)\ln\left(\frac{I_o}{I_t}\right)$$

where $D$ is the thickness of the samples and $I_o$ and $I_t$ are the corresponding intensities of incident and the transmitted radiations respectively.

### FTIR, Raman and NMR spectroscopy

The IR absorption spectra were measured to study the structural behavior of the glass samples over a specific frequency range 400-1400 cm$^{-1}$ obtained by employing potassium bromide (KBr) pallet technique of about 1:100 ratio to form a thin pellet. The pressure gauge of 5 tons cm$^{-2}$ for five minutes to yield a transparent pallet of 0.5 mm thickness ready for mounting in the IR spectrophotometer (Perkin-Elmer 1710 FT-IR spectrophotometer). Raman spectra of the samples was still measured at room temperature with Raman X-plora spectrometer over a wave number range 200-2000 cm$^{-1}$ and Neodymium Yttrium Aluminum Garnet (Nd:YAG) as laser source for 600-1200 nm grating. The solid-state $^{31}$P MAS NMR spectra (Figure 4) were obtained by Brucker Advance III HD at high-resolution power of 400 MHz (magnetic field 9.4 T), 16 $\mu$s pulse at recycle time of 2.5 sec with spinning speed of 14 kHz, at constant frequency of phosphorus 161.94 MHz. The entire chemical shift was expressed in parts-per-million (ppm) relative to 85% aqueous phosphate acid.
RESULTS AND DISCUSSION

Structural properties

-X-Ray Diffraction Analysis

The X-ray diffraction pattern of the glass samples 30MgSO$_4$–(70–x) P$_2$O$_5$–xSm$_2$O$_3$ in mol % (Figure 1), there is no sign of devitrification tendencies in the glass system, this indicate that the samples are amorphous in nature, from the spectra obtained there is absence of sharp peaks, it’s lacks long-range order as presented in the previous literature [10, 11].

![Figure 1 XRD pattern of MSP; Sm glasses](image)

-Infrared Spectra Analysis

The spectra (IR) of 30MgSO$_4$–(70–x) P$_2$O$_5$–xSm$_2$O$_3$, with 0.2 ≤ x ≤ 1.0 mol% glasses as shown in Figure 2, the common features of the IR spectra of γ-P$_2$O$_5$ glass are vibrations around 1640 cm$^{-1}$ assigned to bending vibration of hydroxyl group [25, 26] a partially bound water. The characteristic feature for stretching vibration of terminal bond (P=O) at about 1274 cm$^{-1}$ [25, 27] are in Q$^3$ region, the vibration around 1083 cm$^{-1}$ assigned to (PO$_3$)$^2$ in the meta-phosphate region [25,28], the region at about 860–940 cm$^{-1}$ were assigned to asymmetric stretching of P-O-P link, $\nu_m$ (P-O-P) [27,28] in Q$^2$ region, while the band at about 750-810 cm$^{-1}$ have been identified as the symmetric stretching modes of P-O-P units of bridging oxygen link [25, 26] in Q$^2$ chain, the observed band at 514 cm$^{-1}$ attributed to bending modes of δ(PO) group [29,30].
-Raman Spectra Analysis

Figure 3 shows the Raman spectra of phosphate based glass samples in the region of 200 to 1500 cm\(^{-1}\), the shape of the Raman Spectra goes along with those found in literature, exhibiting various phases of symmetric (\(v_s\)) and asymmetric (\(v_a\)) vibrational modes due to predominant covalence of PO\(_4\), PO\(_3\), PO\(_2\) and P-O \([31]\) group, small positions of bands in sulfate overlapped with phosphate units demonstrating tendencies of inter cluster covalent-ionic interactions between phosphate and sulfate groups. In both spectral responses (Raman and IR) on phosphate network activities are based on bending modes, \(v \sim 500\) cm\(^{-1}\), NBOs modes, \(v \sim 940 – 1380\) cm\(^{-1}\) and bridging oxygen modes, \(v \sim 700 – 900\) cm\(^{-1}\)\([32]\). The structural changes were observed as a result of modifier addition in the phosphate glass matrix and the characteristic features of the host network are basically restricted to the higher wave number positions of the spectra (300–1400 cm\(^{-1}\)) and are corresponding to the vibration of anions, the vibrational modes around 340 cm\(^{-1}\) and 530 cm\(^{-1}\) are due to bending modes of O-P-O units, PO\(_2\) torsionional (\(\epsilon\)) vibrations \([29, 33]\), the bands at \(\sim 750\) assigned to symmetric stretching vibrations (\(v_s\)) of P-O-P in PO\(_4\) units \([30, 33]\) and \(v_{as}\) (P-O-P) at about 915 cm\(^{-1}\) \([31]\) in orthophosphate region, the vibrational bands around 1170 cm\(^{-1}\) were attributed to symmetric stretch (\(v_s\)) mode of (PO\(_2\))\(^+\) units in PO\(_4\) tetrahedra \([29, 32]\) and the stretching mode of terminal oxygen (P=O) somewhere in the region \(\sim 1280-1380\) cm\(^{-1}\) ultra-phosphate units \([29, 32]\). The materials compositions and O/P ratio were tabulated in Table 1.
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-NMR Spectra Analysis

The phosphorus-31 NMR spectra (Figure 4) from different Q-value distribution, two spectral lines were recorded, the isotropic resonance in relation to spinning side-bands signifying the anisotropy of chemical shift. The isotropic peak symbolized by $Q^n$ groups with respect to the chemical shift. The isotropic resonance within the ultra-phosphate ($Q^3$) region is about $-45$ ppm, with increase in modifier content the chemical shift in the range of $-20$ to $-30$ ppm for meta-phosphate units, is in good agreement with the previous NMR finding [33, 34], 2 clear isotropic contributions are assigned to $Q^1,Q^2$ units. The spectra are almost similar in shape, with systematic trend move to lower field (taken as the maximum peak) as a result of increase in modifier content. Positive chemical shift are correspond to lower field (downfield) strength while the negative correspond to higher field (up-field) strength. These intermediate bands intensity of the spectra correspond to ultra-, meta- and polyphosphate, other less intense bands occurred in both sides are the spinning side-bands indicated with asterisks, it happens at a result of anisotropies of phosphate species which signifies the anisotropy of chemical shift (CSA). The isotropic shift are indicated by $Q^n$ distribution with respect to the chemical shift (n represent the number of BO per PO$_4$ tetrahedra) [35].

![Figure 3 Raman spectra for MSP; Sm glasses](image)

![Figure 4 $^{31}$P NMR spectra for MSP; Sm glasses](image)
Table 1: Materials composition and O/P ratio

<table>
<thead>
<tr>
<th>Sample codes</th>
<th>MgSO₄</th>
<th>P₂O₅</th>
<th>Sm₂O₃</th>
<th>O/P</th>
</tr>
</thead>
<tbody>
<tr>
<td>MSPSm 1</td>
<td>30</td>
<td>69.8</td>
<td>0.2</td>
<td>3.364</td>
</tr>
<tr>
<td>MSPSm 2</td>
<td>30</td>
<td>69.6</td>
<td>0.4</td>
<td>3.370</td>
</tr>
<tr>
<td>MSPSm 3</td>
<td>30</td>
<td>69.4</td>
<td>0.6</td>
<td>3.377</td>
</tr>
<tr>
<td>MSPSm 4</td>
<td>30</td>
<td>69.2</td>
<td>0.8</td>
<td>3.384</td>
</tr>
<tr>
<td>MSPSm 5</td>
<td>30</td>
<td>69.0</td>
<td>1.0</td>
<td>3.391</td>
</tr>
</tbody>
</table>

Physical properties

Density (ρ) is the physical properties used to determine the degree of compactivity of materials which are related to the change in geometrical configurations of the sample [36], other physical properties were obtained using molecular weights and density from a suitable expressions such as molar volume (Vₘ), ion concentration, inter nuclear distance et cetera, the density and refractive index formula as indicated in equations (1 and 2) above. We obtain the rare earth ions concentrations (Nᵢ) as well as the polaron radius (rᵢ) from following expressions (4 and 5).

\[ Nᵢ = \frac{x \rho N_A}{M_T} \]  
\[ rᵢ (Å) = \frac{1}{2} \left( \frac{\pi}{6 Nᵢ} \right)^{1/3} \]

where \( N_A \) is the Avogadro’s number, \( x \) is the mole fraction of the rare earth ions and \( M_T \) is the average molar weight of the samples, the physical properties were tabulated in Table 2. Figure 5 demonstrate the dependence of density and molar volume of the samples, it shows that the density and molar volume changes with increase in Magnesium sulfate content, the trend of the molar volume were directly depends on the density of glasses. There is a decrease in polaron radius with increase in ion concentration (Figure 6), the result found is consistent with [37, 38] inter-nuclear distance decreases with increase in mol % of samarium ions content.

Table 2: Physical properties of the Sm³⁺ doped Magnesium sulfate ultra-phosphate glasses

<table>
<thead>
<tr>
<th>Physical measurements</th>
<th>Samples</th>
</tr>
</thead>
<tbody>
<tr>
<td>Av. molar weight</td>
<td>0.2%</td>
</tr>
<tr>
<td>M (g)</td>
<td>102.108</td>
</tr>
<tr>
<td>Density (g/cm³)</td>
<td>2.3657</td>
</tr>
<tr>
<td>Molar volume (cm³)</td>
<td>43.162</td>
</tr>
<tr>
<td>Sm³⁺ ion concentration (× 10²¹ ion/cm³)</td>
<td>2.789</td>
</tr>
<tr>
<td>Polaron radius (Å × 10⁻⁸)</td>
<td>2.863</td>
</tr>
<tr>
<td>Inter-nuclear distance (Å × 10⁻⁸)</td>
<td>7.104</td>
</tr>
<tr>
<td>Field strength (× 10¹⁷ /cm²)</td>
<td>0.756</td>
</tr>
<tr>
<td>Refractive index</td>
<td>2.159</td>
</tr>
<tr>
<td>Dielectric constant</td>
<td>4.661</td>
</tr>
</tbody>
</table>
Molar refractivity \( R_m (\text{cm}^{-3}) \) 23.723 23.910 23.858 23.811 23.716
Molar polarizability \( \alpha_m \times 10^{-24} \) 9.405 9.479 9.456 9.459 9.472
Direct mobility gap (eV) \( E_{\text{dir}} \) (eV) 4.052 3.941 3.932 3.907 3.886
Indirect mobility gap (eV) \( E_{\text{indir}} \) (eV) 4.514 4.538 4.544 4.574 4.512
Width of tail \( \Delta E \) (eV) 0.306 0.291 0.224 0.232 0.210

| Figure 5 | Variation of density against molar volume with increase in mol % of Sm\(_2\)O\(_3\) ion. |
| Figure 6 | Variation of N (ions/cm) and polaron radius with increase in mol % of Sm\(_2\)O\(_3\) ion |

**Optical absorption spectra**

The measured optical absorption spectra for 30MgSO\(_4\)-(70-x) P\(_2\)O\(_5\)-xSm\(_2\)O\(_3\) glasses in the wave length range 350-1800 nm were indicated in Figure 7. The bands were obtained in both glasses as they differ with their intensities. Nine prominent transitions peaks appeared in the UV-Vis-NIR region corresponding to \(^6\)H\(_{5/2}\)→\(^6\)P\(_{3/2}\), \(^6\)H\(_{5/2}\)→\(^4\)I\(_{11/2}\), \(^6\)H\(_{5/2}\)→\(^4\)F\(_{11/2}\), \(^6\)H\(_{5/2}\)→\(^4\)F\(_{7/2}\), \(^6\)H\(_{5/2}\)→\(^4\)T\(_{7/2}\), \(^6\)H\(_{5/2}\)→\(^4\)T\(_{5/2}\), \(^6\)H\(_{5/2}\)→\(^4\)T\(_{3/2}\), \(^6\)H\(_{5/2}\)→\(^4\)H\(_{15/2}\) and \(^6\)H\(_{5/2}\)→\(^4\)F\(_{15/2}\) transition at 400, 472, 945, 1080, 1234, 1380, 1481, 1544 and 1589 nm respectively, the peaks observed are analogous with the other findings [15,39] and the idea of transitions are correspond to carnell et al [40]. The spectral line is a clear characteristics of 4f–4f
The absorption of trivalent $\text{Sm}_2\text{O}_3$ corresponds to $^6\text{H}_{5/2}$ ground state to an excited states transitions with the energy band positions, the intensities observed changes directly with dopant ($\text{Sm}^{3+}$) concentration where $^6\text{H}_{5/2} \rightarrow ^4\text{F}_{7/2}$ act as the hyper-sensitive transition, the ground state absorption emerge due to ED contributions by obeying the selection rule $|\Delta S| = 0, |\Delta L| \leq 2, \text{and} |\Delta J| \leq 2$ [41].

![Figure 7 Absorption spectra for 30MgSO$_4$-(70-x) $\text{P}_2\text{O}_5$-x$\text{Sm}_2\text{O}_3$, with 0.2 $\leq x \leq$ 1.0 mol% glasses](image)

**Band gap energy ($\Delta E$) and Urbach's analysis**

The optical band gap values of magnesium sulfate ultra-phosphate glass were obtained through absorption spectra, having two transitions (direct and indirect). From the absorption spectral region, Davis and Mott come up with the following relation for direct and indirect optical transitions is given by

$$\alpha(v) = N(hv - E_{op}^g)^n / hv$$  \hspace{1cm} (6)

where the exponent in (6) as $n = 2$ (indirect allowed) and $\frac{1}{2}$ for allowed direct transitions respectively, $N$ is energy independent constant (band tailing parameter), $E_{op}^g$ is optical band gap of $hv$ as photon energy of incident radiation (Tauc’s gap), the scheme of $(\alpha \cdot v)^{1/2} = 0$ and $(\alpha \cdot v)^{2} = 0$ for indirect and the direct allowed transitions [42]. The $E_{op}^g$ energies can be obtained from (6) by extrapolating the absorption coefficient to zero absorption, the dependency of $(\alpha \cdot v)^{2}$ on $hv$ have been plotted as in Figure 6(a). The $E_{op}^g$ value decreases with increase in samarium ion content, where the modifier addition contribute to the change in phosphate structure, this may also lead to decrease in optical band gap [43] the increase in NBOs indicates the upturn of free electrons and result in the decreasing number of optical band gap energy [44]. It was discovered that the increase in dopant content may improve the degree of localization there by creating defects in the charge distribution; as such, the energy of the nearest oxygen ions were induced to the top valence band to raised the number of donor centers of the system. The upward turn in donor centers causes the downturn in the band gap energy values [45]. The Urbach energy ($\Delta E$) provides an information about the disorderliness between the host and impurities atoms, when high energy UV light is absorbed by the glassy phases, the values obtained were evaluated by taking the inverse of the
slopes of ln α versus photon energy as presented in Figure 6(b), the band tails are represented by a band tail parameter, ΔE, (Urbach energy) given by [40]

\[ \alpha(v) = \beta \exp \left( \frac{hv}{\Delta E} \right) \]  

where \( \beta \) is constant referred to band tailing parameter and ΔE is the Urbach energy interpreted as the width of the localized tail state in the optical band gap, the urbach energy values were listed in Table 2.

**DISCUSSION**

Phosphate based glasses (PBGs) constitute of an inorganic phosphate network (as \( P_4^{3-} \) tetrahedral units serves as main building block). In \( v \)-Phosphate, the phosphorus (P) and oxygen (O) atoms are configured in 3-D polymerized network structure of rings and chains, the tertahedra is link to adjacent units by 3 of 4 vertices where the fourth side is occupied by double-bond oxygen atom [46]. The tetrahedra \( (P_4^{3-}) \) are formed by \( sp^3 \) hybridization of P outer-shell electrons (as \( 3s^2 \ 3p^3 \)), this tetrahedra units are expressed in terms on \( Q^n \) nomenclature (n represent the number of BOs per phosphate tetrahedra. The network structure is strongly depends on the modifier oxide content, the condensed phosphate \( Q^n \) groups are ultraphosphate, \( Q^3 \) (branching unit), meta-phosphate, \( Q^2 \) (middle unit), poly-phosphate, \( Q^1 \) (end unit) and ortho-phosphate unit, \( Q^0 \) (isolated tetrahedra), the \( Q^3 \) (branching unit) react with modifier oxides to form \( Q^2 \) (middle unit), all \( Q \) species depend on the O/P ratio in accordance to stoichiometry refer to Table 1. Incorporation of modifier oxide (MgO) to phosphate effect the phosphate link to formed P-O-K (K= metal ion), hence de-polymerized the structure. The quantitative measurements of \( Q^n \) distribution are in consistent with the Van Wazar predictions for the structure of phosphate [47]. Van Wazar formalized the transformation of \( P_2O_5 \) structures with increase or decrease in compositions. The embedded modifier oxides (MgO in %) can be schematically represented by modifier pseudo-reaction according to \( 2Q^3 + R_2O \rightarrow Q^{n-1} \) where \( R_2O \) or RO are the monovalent or divalent cations. In terms \( Q^3 \) branching units the modifier react to form \( Q^2 \) species (as \( 2Q^3 + R_2O \rightarrow 2Q^2 \)) hence, the number of NBOs increases. The degree of
phosphate polymerizations were express by the relative amount of modifier content which described in term of elementary structural units ultra-, $Q^1$ (POO$_{1/2}$)$_0$, meta-$Q^2$ (POO$_{2/2}$O)$_1$, poly-$Q^3$ (POO$_{3/2}$O)$_2$ and ortho-phosphate, $Q^0$ (POO$_{0/2}$O)$_3$ respectively. The interaction of [POO$_{2/2}$O]$_n$ ions with sulfate to form $SPO_3^-$ (dithiophosphate) species [10, 47]. The small fraction of phosphate tetrahedra (PO$_4$) may associate with ring structures which limit the chain length [33], but the development of phosphate rings does not affect the $Q^n$ distribution. It is worth considering that, the atomic arrangements in the short and intermediate range of networks are the most significant factor that determines the emission feature of the materials. The covalency/or symmetry of rare earth ions are different when modifier oxides are merge between the longer chain of RE ions in S-O-P network. Likewise, the differences in the structure of phosphate and sulfate ions with respect to their concentration change the crystal field around the lanthanides ions in the samples [48]. The relative MgO content cannot be obtain by the NMR spectra, however, the region of $Q^n$ distribution of ultra-, meta- and orthophosphates region with their ratio remain valid [34] in both FTIR, Raman and NMR spectra.

**CONCLUSION**

The average order of phosphate chains in the magnesium sulfate ultra-phosphate glasses becomes increasingly shorter as the modifier (MgSO$_4$) content is gradually increased. The process is quantitatively expressed by the increasing concentration of pyro-phosphate ($Q^1$) and ortho-phosphate ($Q^0$) groups determined by FTIR and Raman spectra with different degrees of polymerization were easily resolve by NMR in $Q^1$, $Q^2$ and $Q^3$ units. The experiments indicate the progressive de-polymerization of phosphate structure, the structure of the glass samples becomes more compressed, and this attributed to the increasing connectivity of MgSO$_4$ and the samarium ions content by sharing the NBOs atoms leading to the reduction of the space distances. The increasing trend of $\rho$ (density) due to increase in Sm$^{3+}$ oxide content with decrease in molar volume result in decrease in the bond length on the inter atomic spacing in between the atoms. The refractive index (RI) increases with increase of dopant owing to decrease in optical energy gap. From the optical analysis, we concluded that upturn of samarium oxides results in decrease of optical energy gap attributed to increase in NBOs ions. The absorption spectra affirm that the fundamental absorption edge shifts to higher wavelength as the dopant content increases. Optical band gap (direct and indirect) and Urbach are found in ranges, the dependence of $\alpha (v)$ on $hv$ obeys the Urbach rule. Our studies confirmed the stability and suitability of the materials in the progress of optical laser materials.

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